# INVESTIGATION ON MECHANOCHEMICAL DEHYDRATION OF $\gamma$ -FeOOH

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The process of mechanochemical dehydration of  $\gamma$ -FeOOH has been investigated. It has been established that grinding of  $\gamma$ -FeOOH leads to its direct transformation into  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The IR spectroscopy and X-ray phase analyses did not detect  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> in the dehydration products. During dehydration,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with a defective magnetic structure is formed. The smallest size of the hematite particles obtained is 10 nm. The difference between mechanochemical and controlled thermal dehydration of  $\gamma$ -FeOOH is discussed.

The mechanochemical dehydration of metal hydroxides has been the subject of a series of investigations<sup>1-4</sup>. The cases in which oxides with different phase compositions are obtained are especially interesting. This is the case of  $\gamma$ -FeOOH. It may be decomposed quantitatively to  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> or  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, depending on the conditions of thermal treatment. Some preliminary results in this respect have already been published<sup>5,6</sup>. The present paper contains data from a study of the behaviour of  $\gamma$ -FeOOH during the grinding process.

# EXPERIMENTAL

The initial  $\gamma$ -FeOOH was obtained by oxidative precipitation from an FeSO<sub>4</sub> solution<sup>7</sup>. The mechanochemical treatment consisted in grinding the lepidocrocite in an automatic agate mortar. The IR spectra were obtained with Beckmann IR and Specord UR-20 spectrometers. The X-ray analysis was carried out on a DRON-1 apparatus using CuK<sub>a</sub> radiation. The Moessbauer spectra were taken on an NTA-1024 instrument at room temperature. The specific surface areas were measured by a modified BET method. The products were reduced at a hydrogen pressure of 0.6 Pa in a gradientless regime<sup>8</sup>.

# **RESULTS AND DISCUSSION**

Electron microscopy data showed that the morphology of the  $\gamma$ -FeOOH particles changed already at the beginning of the grinding process when dehydration was observed.

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Fig. 1 shows a gradual disappearance of the lines of  $\gamma$ -FeOOH with increasing duration of the grinding process and appearance of lines of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. No  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> was detected in the solid reaction products either by X-ray phase analysis and the IR spectroscopy. The results from both analyses indicate complete dehydration of the initial  $\gamma$ -FeOOH to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> after 80 h grinding (Fig. 1).

Moessbauer studies showed that as a result of mechanochemical dehydration of  $\gamma$ -FeOOH,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with a defective magnetic structure quite different from that of standard  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> was obtained. A strong decrease in the value of the internal magnetic field was observed ( $H_{eff} = 462$  kOe for a sample ground for 57 h) as well as Non-Lorenzian shape of the Moessbauer spectra, and asymmetry of the central doublet.

The standard  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> has an internal magnetic field of 520 kOe. The deviations may be attributed to defects appearing during the sample grinding. This explanation is based on: 1) a small difference in the specific surface areas of the initial  $\gamma$ -FeOOH and the samples obtained (Table I); 2) the fact that the sample particles are not small enough to lead to superparamagnetism, which is evident from the strong X-ray lines in the diagram (Fig. 1). During grinding of iron oxides and ferrites a similar significant change in the Moessbauer spectra, *e.g.* a decrease in the value of the internal magnetic field and asymmetry have been observed<sup>9-12</sup>. These changes could be attributed to defects appearing during the grinding process<sup>9-12</sup>.

The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> obtained after 80 h grinding has fewer defects in its magnetic structure. Its Moessbauer spectrum has six lines. The value of the internal magnetic field increases ( $H_{eff} = 482 \text{ kOe}$ ). This value is still considerably lower than those of the internal magnetic field of the standard  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and the hematite obtained by controlled thermal decomposition. The ratio of the peak intensities of the oxide obtained after 80 h grinding is quite different from that for standard  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. There is also a significant difference between this ratio for hematite obtained by mechanochemical decomposition and that for hematite resulting from controlled thermal decomposition even when both compounds have close specific surface areas. These results show that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> obtained by grinding has more defects in its structure than the hematite prepared by conventional methods.

A comparative study was carried out on the reactivities of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(*I*) obtained by grinding of  $\gamma$ -FeOOH for 80 h and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(*II*) prepared by controlled thermal decomposition of  $\gamma$ -FeOOH. Both oxides had very close specific surface areas: 115 m<sup>2</sup> g<sup>-1</sup> for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(*I*) and 121 m<sup>2</sup> g<sup>-1</sup> for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(*II*). It was established that  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(*I*) was reduced to Fe<sub>3</sub>O<sub>4</sub> much more rapidly than  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(*II*) (Fig. 2). This indicates that  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(*I*) is more reactive on account of its defective structure.

The specific surface areas of the samples obtained during grinding showed a small change (Table I) due to reaching the limit size of 10 nm when the processes of particle aggregation are intense<sup>4,9,13,14</sup>.

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The above investigations showed that  $\gamma$ -FeOOH is dehydrated during its grinding, resulting in the formation of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and the appearance of significant defects in the structure of the solid reaction products.

# TABLE I

Dependence of specific surface area  $(m^2 g^{-1})$  changes on the duration (h) of mechanical treatment of  $\gamma$ -FeOOH<sup>a</sup>

Time, h	25	33	49	57	80	100	
Specific surface area	97	103	100	107	115	105	

<sup>*a*</sup> Specific surface area of initial  $\gamma$ -FeOOH = 115 m<sup>2</sup> g<sup>-1</sup>.



Fig. 1

X-ray phase diagram of  $\gamma$ -FeOOH 1 and products of its mechanochemical dehydration. Samples 2, 3, 4, 5 ground for 25, 49, 57, and 80 h. ----  $\gamma$ -FeOOH, -----  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>



# Fig. 2

Change of hydrogen pressure in the calibrated volume of a static-circulation apparatus  $(\Delta P)$  during the reduction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> to Fe<sub>3</sub>O<sub>4</sub>. 1 Reduction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (I) obtained by mechanochemical decomposition of  $\gamma$ -FeOOH, 2 reduction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (II) obtained by controlled thermal decomposition of  $\gamma$ -FeOOH

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During the mechanochemical dehydration of  $\gamma$ -FeOOH, no  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>, which is characteristic of controlled thermal decomposition of lepidocrocite, is observed. This may be ascribed to peculiarities of mechanochemical reactions, *e.g.* to a local sharp increase of temperature during a mechanical treatment<sup>12,15</sup>, which favours the conversion of  $\gamma$ -FeOOH to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. This process can also be accompanied by the liberation of the energy accumulated in the defects<sup>4,16</sup>. We are of the opinion that the local sharp increase of temperature is prevailing. Thus, it may be expected that grinding in a heat-absorbing medium would lead to the formation of reaction products with a different composition.

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